# INFLUENCE OF HYDROGEN-THERMAL TREATMENT OF NITRIDE POWDERS ON THE PROPERTIES OF COMPOSITES ON THEIR BASE

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## Introduction

Method of hydrogen-thermal treatment (HTT) of nonmetallic nitrides powders (BN, AlN,  $Si_3N_4$ ) has been developed at the IPMS. HTT purifies powders from admixtures, allows both to decrease temperature, time of sintering and to obtain sintered specimens from these powders with higher density, strength and specific electric resistance [1, 2].

#### Results and discussion

The aim of the work is investigation of influence of HTT of non-metallic refractory compounds on the properties of composites based on these powders. Thermal treatment of powders at 1000-1200 °C in hydrogen with the use of catalyst promotes the purification from oxygen, changing of their granulometric composition due to destruction of filmlike connections among particles in aggregates and destruction of these aggregates.

Sintered powder specimens after HTT have smaller average grain size, higher content of anisometric grains and contain high-density formations from dispersed grains.

To investigate the influence of HTT on composites properties in  $\alpha$ -Si<sub>3</sub>N<sub>4</sub>-AlN, AlN-BN and BN-B<sub>4</sub>C systems green-body mixtures and mixtures of treated powders have been prepared. Above mention mixtures have been treated with hydrogen at 1100 °C during 30 min. Hydrogen has been obtained by decomposition of TiH<sub>2</sub>. Obtained powder mixtures have been pressed and sintered in nitrogen at 1550-1900 °C.

Pressed  $\alpha$ -Si<sub>3</sub>N<sub>4</sub>-AlN powders have been sintered at 1550 °C. Al powder as an addition to  $\alpha$ -Si<sub>3</sub>N<sub>4</sub> was expected to be nitrided during the sintering and form AlN phase. Phase composition of specimens and their strength properties are shown in table 1.

It was shown by X-Ray analysis that in specimens obtained from HTT treated powders Si<sub>2</sub>ON admixture disappears and lines of Si<sub>3</sub>N<sub>4</sub> and AlN became clearer. It evidences the ordering of crystal structures. Herein strength properties of specimens increase. Electric resistance of all the specimens exceeds 10 Ohm·cm (20 °C).

Pressed powders of aluminium and boron nitrides were sintered at 1900 °C. Properties of the specimens are shown in table 2. As follows from the data sintered specimens obtained from treated powders have higher density, compression strength, heat resistance and electric resistance at heating. Coefficient of thermal expansion (CTE) and phase composition do not change. In initial specimens at small angles besides of two main phases there is a small amount of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> admixtures which does not disappear after HTT, while the lines of main phases became clearer.

BN-B<sub>4</sub>C specimens were sintered at 1900 °C. Sintered specimens data are shown in table 3. It is evident from table 3, that due to HTT of green body powders the electric resistances at heating and compression strength of sintered specimens are higher. Herein coefficient of thermal expansion decreases a little. There are BN, B<sub>4</sub>C and B<sub>2</sub>O<sub>3</sub> phases in initial specimens and in treated specimens only two phases left as there is no B<sub>2</sub>O<sub>3</sub>.

# **Conclusions**

Preliminary date of influence of HTT of mixtures of initial powders on  $\alpha$ -Si<sub>3</sub>N<sub>4</sub>-AlN, AlN-BN and BN-B<sub>4</sub>C composites properties have been obtained.

It was determined that HTT of initial powders increases the density of sintered specimens their strength, electric resistance at heating, heat resistance, puts crystal structure in order, reduces quantity of oxygen-containing admixtures.

Table 1 – Properties of AlN-Si<sub>3</sub>N<sub>4</sub> system materials

	Addition	Property				
Treatment	to $\alpha$ -Si <sub>3</sub> N <sub>4</sub> ,	Densi-	Compression	Thermal shocks	Phase composition	
	mas. %, (init.	ty,	strength	(1800-20 °C,		
	mixture)	g/cm <sup>3</sup>	σ <sub>compr.</sub> , Mpa	air)		
Untreated	20 AlN	2,58	158	39	$\alpha$ -Si <sub>3</sub> N <sub>4</sub> , AlN, $\beta$ -Si <sub>3</sub> N <sub>4</sub>	
	20 Al	2,62	167	45	$\alpha$ -Si <sub>3</sub> N <sub>4</sub> , AlN, $\beta$ -Si <sub>3</sub> N <sub>4</sub> ,	
	(powder)				${ m Si_2ON}$	
HTT	20 AlN	2,76	169	43	$\alpha$ -Si <sub>3</sub> N <sub>4</sub> , AlN, $\beta$ -Si <sub>3</sub> N <sub>4</sub>	
treated						
	20 Al	2,80	171	49	$\alpha$ -Si <sub>3</sub> N <sub>4</sub> , AlN, $\beta$ -Si <sub>3</sub> N <sub>4</sub>	
	_ (powder)				•	

Table 2 – Properties of AIN-BN system materials (mass composition 1:1)

Treatment	Property						
	Density,	Compression	Thermal shocks	Specific electric resis-	Phase composi-		
	g/cm <sup>3</sup>	strength $\sigma_{compr.}$	(1800-20 °C, air)	tance, Ohm·cm (t, °C)	tion		
		MPa					
Untreated	2,17	90,1	88	$10^{12}(20); 2\cdot10^{9}(500);$	AlN, BN,		
				$5.10^6 (1000)$	$\alpha$ -Al <sub>2</sub> O <sub>3</sub> (traces)		
HTT treated	2,35	102,3	105	$10^{12}(20); 1,8\cdot10^{9}(500);$	AlN, BN,		
				$6,2\cdot10^{7}$ (1000)	$\alpha$ -Al <sub>2</sub> O <sub>3</sub> (traces)		

Table 3 – Properties of BN-B<sub>4</sub>C system materials (mass composition 1:1)

Treatment	Property							
	Density,	Compression strength	Specific electric resis-	Phase composition				
	g/cm <sup>3</sup>	σ <sub>compr.</sub> , MPa	tance, Ohm·cm (t, °C)					
Untreated	1,9	15,9	$10^{13}(20); 1,5\cdot10^7$	BN, $B_4C$ , $B_2O_3$				
			$(1000); 2 \cdot 10^4 (1500)$					
HTT treated	2,1	29,8	$10^{13}(20); 1,6\cdot10^9$	BN, B <sub>4</sub> C				
			$(1000); 1 \cdot 10^6 (1500)$					

## References

- 1. Morozova R.A., Morozov I.A., Dubovik T.V. et al. The influence of hydrogen-thermal treatment on the densification and structure of nitrides at sintering, *Hydrogen Materials Science and Chemistry of Metal Hydrides* (ICHMS'2001), 243-258, NATO Science Series, II. Maths., Phys. Chem., Vol. 82, Kluwer, Dordrecht 2002.
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