## INFLUENCE OF HYDROGEN ON MAGNETOCRYSTALLINE ANISOTROPY OF TbFe<sub>6</sub>Co<sub>5</sub>Ti SINGLE CRYSTAL

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## Introduction

The magnetic properties of iron-rich rare-earth intermetallic RFe<sub>11</sub>Ti compounds have attracted much attention in recent years [1-2]. This series of alloys adopts the tetragonal ThMn<sub>12</sub>-type crystal structure (space group I4/mmm). The unit cell contains 26 atoms (two formula units per unit cell). The rare-earth atoms occupy only one kind of high-symmetry site (2a) while iron and titanium atoms occupy three different sites (8i, 8j, and 8f). The RFe<sub>11</sub>Ti compounds have high Curie temperatures  $(T_C)$  around 600 K. Some of them have a large magnetic anisotropy and high saturation magnetization and may be considered as candidates for permanent magnet potential materials.

A replace of Fe by Co leads to increase the Curie temperature and a change of magnetocrystalline anisotropy of the 3dsublattice [3-4]. The easy magnetization direction coincides with [100] axis for TbFe<sub>11</sub>Ti at 4.2 K [3]. As the temperature is increased a first order spinreorientation transition (SRT) occurs at 325 K. where the easy direction changes from the [100] to the [001] axis. The introduction of hydrogen into the crystalline lattice of TbFe<sub>11</sub>Ti induces easy plane states at temperature below T<sub>C</sub> [5]. At Fe replacement by Co temperature of SRT decrease [4]. The purpose of this work was to investigate the magnetic properties (magnetization and magnetic anisotropy) of the of the TbFe<sub>6</sub>Co<sub>5</sub>Ti compound and its hydride. In contrast to previous works we investigate single crystals samples of hydride.

## Results and discussion

The detailed description of the preparation method of the single crystal samples TbFe<sub>11-x</sub>Co<sub>x</sub>Ti used in the works were given somewhere else [4]. The metal hydride syntheses were carried out in a high-pressure reactor chamber of a conventional Sieverts-type volumetric system. The samples were activated for 4 hours in vacuum (4•10<sup>-4</sup> Pa) at 670 K. At this temperature high purity hydrogen

gas obtained from the  $LaNi_5H_6$  hydrogen storage was admitted at a pressure of 1.2 MPa. To achieve a good homogenization, the samples were slowly cooled (about 4 K per hour) down to room temperature. The amount of absorbed hydrogen was determined from the hydrogen pressure change in the reactor chamber.

Magnetization measurements were carried out with a SQUID (Quantum Design MPMS 5-S) magnetometer from 5 to 300 K in magnetic fields up to 50 kOe.

X-ray diffraction measurements show that the ThMn<sub>12</sub>-type crystalline structure of TbFe<sub>6</sub>Co<sub>5</sub>Ti compound is retained upon hydrogenation. The

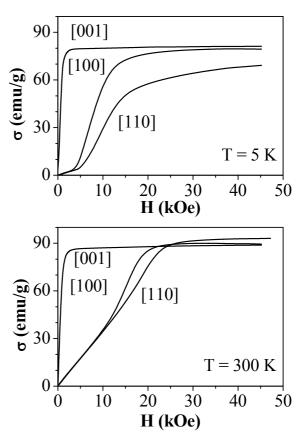


Fig. 1. Isothermal magnetization curves at 5 and 300 K on a single crystal of TbFe<sub>6</sub>Co<sub>5</sub>Ti for magnetic field along the main symmetry directions: [100], [110] and [001].

Compounds	$\sigma_{\rm S}$ (emu/g)		$K_1 10^{-7} (erg/cm^3)$		$K_2 10^{-7} (erg/cm^3)$	
	5 K	300 K	5 K	300 K	5 K	300 K
TbFe <sub>6</sub> Co <sub>5</sub> Ti	81	88	2.95	0.74	-2.88	-0.43
TbFe <sub>6</sub> Co <sub>5</sub> TiH	73.7	85.1	-1.83	-0.24	0.73	0.09

introduction of hydrogen atoms is increased the lattice constants. It leads to an isotropic volume expansion with the c/a ratio being almost unchanged.

Figs. 1 and 2 show the isotherms obtained from the magnetization measurements at  $T=5\,\mathrm{K}$  and 300 K for TbFe<sub>6</sub>Co<sub>5</sub>Ti and its hydride, respectively. Unlike TbFe<sub>11</sub>Ti, the TbFe<sub>6</sub>Co<sub>5</sub>Ti single crystal exhibits a uniaxial magnetic anisotropy over the entire studied range of temperatures from 4.2 K to the Curie points. It is show that Fe replacement by Co suppress easy plane anisotropy at Co concentration x=5.

The magnetization curves for the TbFe $_6$ Co $_5$ Ti single crystal along the [110] and [100] directions in the basal plane (Fig. 1) show a sharp jump of the magnetization in the temperature below 300 K at the specific threshold fields H $_{cr}$ . The H $_{cr}$  increases with temperature. These transitions — the first-order magnetization processes (FOMP) [6]. This can be explained by the irreversible rotation of the magnetization vector at H = H $_{cr}$ . Using theory [6]

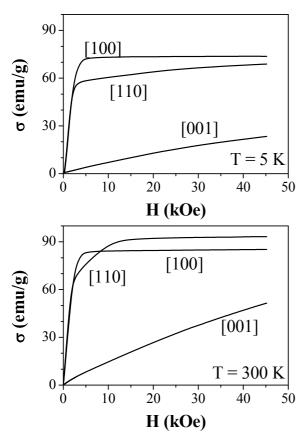


Fig. 2. Isothermal magnetization curves at 5 and 300 K on a single crystal of TbFe<sub>6</sub>Co<sub>5</sub>TiH for magnetic field along the main symmetry directions: [100], [110] and [001].

we calculate anisotropy constant which represent in Table.

As can be seen from Fig. 2, hydrogenation induced easy plane anisotropy in TbFe<sub>6</sub>Co<sub>5</sub>TiH. The [100] becomes easy magnetization axis and critical fields are not observed in the field dependences of the magnetization TbFe<sub>6</sub>Co<sub>5</sub>TiH. As can be seen from Table, the anisotropy constants change sign under hydrogenation. These facts indicate there is a change of magnetic properties in both Tb and 3dsublattices under hydrogenation of TbFe<sub>6</sub>Co<sub>5</sub>Ti [7].

The magnetocrystalline anisotropy depends strongly on the values of the electric fields at the site of the R-ions. The interstitial H atoms occupy the 2b sites adjacent to the rare earth, creating a change of crystal field. The insertion of H leads to an expansion of the temperature range in which the easy plane magnetic anisotropy of the rare-earth sublattice dominates over that of the 3d-sublattice.

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